### Lecture 6: Strengthening of Plain Carbon Steels

#### **MMat 380**

## Contributions to strength in steels

- Base material
- •Solid solution strengthening (i.e., %Mn)
- •Grain size (ferrite)
- Precipitates (distance between ppts)
- Cold work (dislocation density)

#### Steels - Fe-C-Mn alloys

- Mn added as ferromanganese
  - Helps to de-sulpherize steel MnS (Fe-S brittleness)
  - Powerful solid solution strengthener
  - Powerful effect on heat treating med. %C (0.3%C) and high %C steels which are usually Q&T

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AISI-SAE No.	% C	% Mn	
1006	0.08 max.	0.25-0.40	
1010	0.08 - 0.13	0.30 - 0.60	
1015	0.13 - 0.18	0.30 - 0.60	
1020	0.18 - 0.23	0.30 - 0.60	
1025	0.22 - 0.28	0.30 - 0.60	
1030	0.28 - 0.34	0.60 - 0.90	1xxx - %C
1035	0.32 - 0.38	0.60 - 0.90	1 A A A - /0 C
1040	0.37 - 0.44	0.60 - 0.90	in steel
1045	0.43 - 0.50	0.60 - 0.90	111 21661
1050	0.48 - 0.55	0.60-0.90	
1055	0.50 - 0.60	0.60 - 0.90	
1065	0.60 - 0.70	0.60 - 0.90	
1070	0.65 - 0.75	0.60-0.90	
1075	0.70 - 0.80	0.40 - 0.70	
1080	0.75 - 0.88	0.60 - 0.90	
1085	0.80 - 0.93	0.70 - 1.00	
1090	0.85-0.98	0.60 - 0.90	
1095	0.90 - 1.03	0.30-0.50	

P, 0.040 max; S, 0.05 max.

# Strengthening low C steels (0-0.3%C)

#### 2 major ways:

- increase carbon content
  - sacrifice % elongation; toughness because of Fe<sub>3</sub>C
- · decrease grain size
  - increase strength but doesn't affect ductility
  - Hall-Petch Equation  $\sigma_y = \sigma_o + kd^{-1/2}$  d = grain size (mm)
- ∴ better strength without sacrificing ductility and toughness

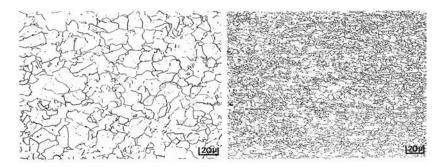
trend: use fine grained steels and lower C content

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#### **Examples**

ASTM	Grain	σ <sub>v</sub> (MPa)	
No		ý	
5	Coarse	250	
8	Fine	300	
12-13 Very fine		500	

## Grain size and strength



335 MPa

540 MPa

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## ASTM grain size

Grain boundaries act as barriers to dislocation motion

 $n=2^{N-1}$ 

 $n = \# grains/in^2 @ 100x mag$ 

N = ASTM grain size No

Note: Table 2.1 pg. 74 Smith

## ASTM grain size

TABLE 2-1 Grain-size number as related to grain count

Timken- ASTM No.		Grains per square inch of image at 100 $\times$			Grains per sq millimeter
		Maximum	Minimum	Mean	(mean actual)
_	3	0.088	0.044	0.06	1
_		0.176	0.088	0.125	2
_		0.35	0.176	0.25	4
	0	0.71	0.35	0.50	8
-	1	1.41	0.71	1.0	16
1	2	2.83	1.41	2.0	32
1	3	5.66	2.83	4.0	64
	4	11.3	5.66	8.0	128
ĺ	5	22.6	11.3	16	256
	6	45.2	22.6	32	512
	7	90.5	45.2	64	1024
1	8	181	90.5	128	2048
_	9	362	181	256	4096
	10	724	362	512	8200
	11	1448	724	1024	16400
	12	2896	1448	2048	32800

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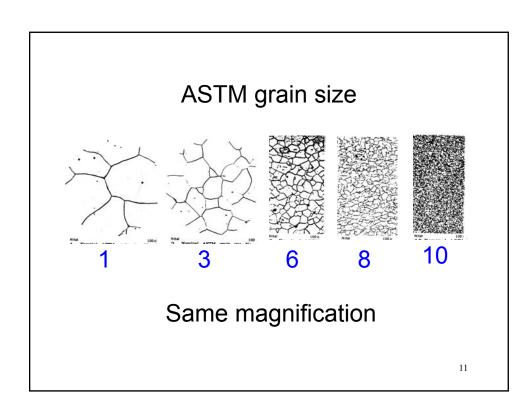
#### Unit conversions

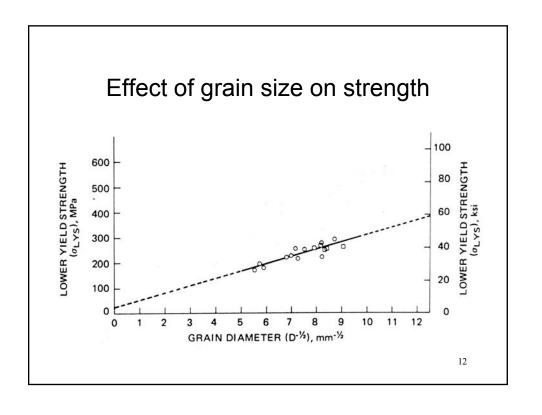
#### ASTM #1

16 grains/mm<sup>2</sup> x  $(25.4)^2$  = 10 323 gr/in<sup>2</sup> = 1.03 gr/in<sup>2</sup> @ 100x

#### ASTM #8

2048 grains/mm<sup>2</sup>x(25.4)  $^2$  =1 321 287 gr/in<sup>2</sup> = 132 gr/in<sup>2</sup> @ 100x not 128





#### **Applications**

Grade	%Mn	Product
1006	0.25-0.4	Sheet
1010-1025	0.3-0.6	Structural
1030	0.6-0.9	Heat treated

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#### Low carbon steels (<0.25%C)

- Sheet Steels
- C <0.1%</li>
  - Cold worked quality surfaces (auto's, appliances)
  - Strain Aging
    - · C, N producing inhomogeneous yielding
    - Need for temper rolling to obtain homogeneous yielding (for smooth surface forming)
- Very low carbon IF Steels (<0.05%C)</li>
  - Paint bake strengthening (+70 MPa Y.S.)

#### Low carbon steels (<0.25%C)

- Structural Steels (<0.15-0.25%C)</li>
- Equilibrium microstructures (25% $\alpha$  +75%P)
- Following hot rolling, use accelerated cooling to decrease γ→ α+P transition temperature (below 721°C)
  - $\therefore$  produce fine  $\alpha$  + increases amount of P
- Ferrite grain size ASTM # 5 (coarse) ASTM # 8 (fine)
- Canadian Standards Association (CSA) CSA G40.21 for quality structural steel

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## High Strength Low Alloy (HSLA) steels

Why low alloy if high alloy provides high strength?

- Traditionally for highest strength in a structural steel the C & Mn levels would be increased
  - i.e. 0.25-0.30%C 1.2-1.5%Mn
  - An increase of 1% Mn will increase YS by ~14%
- This led to **problems** with:
  - Weldability (problem with increased C and Mn)
  - Brittle failure (problem with increased C)
- New approach required: strength but LC
- Now have steels with YS to 550 MPa but with excellent weldability and brittle fracture resistance

#### **HSLA** steels

- Solid solution hardening (Mn)
- Decrease ferrite grain size by
  - Controlled rolling
  - Controlled cooling
- Precipitation hardening
  - Nb (C,N)
  - VC

Typical x70 pipeline steel %C = 0.06; %Mn = 1.50; %Nb and/or V ~0.04 Controlled rolling to produce very fine grain size

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#### Strengthening HSLA steels

- Obtain fine grain size (ASTM 10-13) by:
  - Controlled rolling
  - Controlled cooling
- Can increase yield strength by 100-134 MPa (i.e. 300-440 MPa total)

Element	$\sigma_{\!_{y}}$ increase of $\alpha$ per 1% addition
Cr	6.7
Co	13.4
V	13.4
W	20
Мо	27
Al	40
Ti	67
Ni	80
Mn	94
Si	100
Ве	600
Р	670

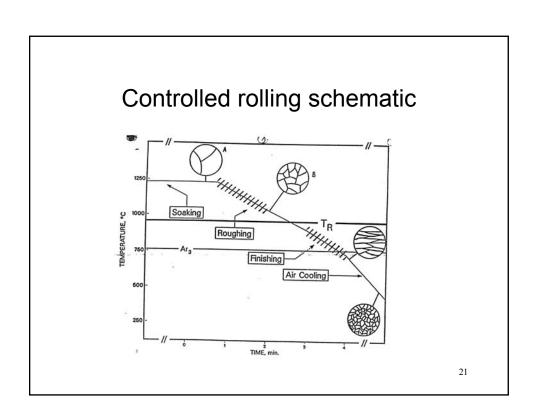
#### Controlled rolling

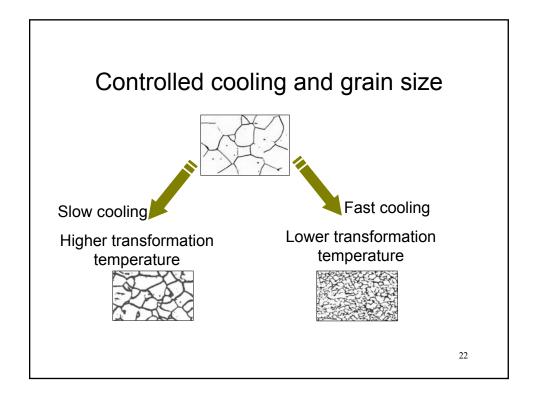
- Normal finish T in hot rolling: 900-1000°C
- Finish at temperatures: 750-800°C
  - Lower temperature
  - R<sub>x</sub> & grain growth after hot rolling takes longer
  - end up with smaller g grain size\ smaller  $\alpha$  grain size

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#### Controlled rolling

- transformation of **deformed**  $\gamma$  gives finest grain size
- $\therefore$  want to **T** of R<sub>x</sub> and make R<sub>x</sub> more difficult
  - Nb in small amounts does this (~0.04%Nb added)
  - Need massive roll force to give required deformation
- Controlled rolled plate typically < 1" in thickness</li>
- Thick plate usually has larger  $\alpha$  grain size because
  - it is finished at a higher T and  $\therefore$  has a larger  $\gamma$  grain size



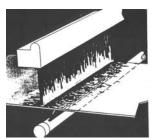


#### Controlled cooling

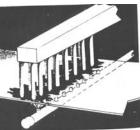
- Now consider cooling rate after rolling
- On increased cooling rate (i.e. H<sub>2</sub>0 jets)
  - $-T_{transf}$  is decreased
  - Higher nucleation rate and low grain growth rate
- When external cooling is the same, a thick plate will cool slower
- $\therefore$  larger  $\alpha$  grain size

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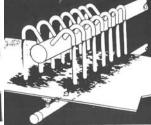
#### Means of controlled cooling



Water-wall strip cooling concept



Dual-jet laminar flow header



Early form of laminar jet flow system – 2 rows of U-tubes from 1 header

#### Thick sections

Why do "thick" sections have lower yield stress?

- When finish rolling at higher T (larger γ grain size) thick sections cool slower
- raises  $T_{transf} \gamma \rightarrow \alpha$
- fewer  $\alpha$  nuclei grow to larger  $\alpha$  grain size  $\therefore$  lower  $\sigma_{_{\! V}}$

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#### Precipitation hardening

Microalloying - Nb, V, Ti

- Nb (C,N) precipitate **during** hot rolling in γ
  - restricts  $\gamma$  grain growth
  - $\boldsymbol{\mathsf{-}}$  refines  $\alpha$  grain size
  - retards R<sub>x</sub> & raises T<sub>transf</sub>
- Pancake grains
  - nuclei closer together therefore finer R<sub>x</sub> grain size
- V VC precipitate on cooling after rolling

#### Strength in HSLA steels

+ Standard C-Mn Steel 200-300 MPa + Decrease grain size 100-134 MPa + Increase Mn 67 MPa + Increase Nb, V, Ti ppt hardening 67-100 MPa

Total: 434-600 MPa

- Can now afford to lower the C content and still have 470-500 MPa steel
- Can have any strength level wanted by varying the degree of strengthening components

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## Effect of austenite grain size

